REPORT DOCUMENTATION PAGE The public reporting burden for this collection of information is estimated to average 1 hour per response, including the time for reviewing instructions, searching existing data sources, gathering and maintaining the data needed, and completing and reviewing the collection of information. Send comments regarding this burden estimate or any other aspect of this collection of information, including suggesstions for reducing this burden, to Washington Headquarters Services, Directorate for Information Operations and Reports, 1215 Jefferson Davis Highway, Suite 1204, Arlington VA, 22202-4302. Respondents should be aware that notwithstanding any other provision of law, no person shall be subject to any oenalty for failing to comply with a collection of information if it does not display a currently valid OMB control number.

PLEASE DO NO	I RETURN YOUR FO	KM TO THE ABOVE	: ADDRESS.				
1. REPORT D	ATE (DD-MM-YY	E (DD-MM-YYYY) 2. REPORT TYPE			3. DATES COVERED (From - To)		
			New Reprint		-		
4. TITLE AN	D SUBTITLE			5a. COl	NTRACT NUMBER		
Effects of pelletization pressure on the physical and chemical				W9111	W911NF-10-1-0076		
properties of the metal–organic frameworks Cu3(BTC)2 and UiO-66			s Cu3(BTC)2 and	5b. GRANT NUMBER			
				5c. PRO 622622	OGRAM ELEMENT NUMBER 2		
6. AUTHORS	<u> </u>			5d. PRC	DJECT NUMBER		
Gregory W. 1	Peterson, Jared B.	DeCoste, T. Grant	Glover, Yougui				
Huang, Himanshu Jasuja, Krista S. Walton			- -	5e. TASK NUMBER			
				5f. WOI	RK UNIT NUMBER		
Georgia Tecl	h Research Corporationsored Programs		ND ADDRESSES		8. PERFORMING ORGANIZATION REPORT NUMBER		
Atlanta, GA		3033	32 -0420				
9. SPONSOR ADDRESS(E	RING/MONITORII S)	NG AGENCY NA	ME(S) AND		10. SPONSOR/MONITOR'S ACRONYM(S) ARO		
U.S. Army Research Office					11. SPONSOR/MONITOR'S REPORT		
P.O. Box 12211 Research Triangle Park, NC 27709-2211				1	NUMBER(S)		
Kesearen III.	aligie I alk, INC 27				57994-CH.25		
	JTION AVAILIBI						
Approved for	public release; dist	ribution is unlimit	ed.				
The views, op		ngs contained in th	nis report are those of the auth designated by other document		not contrued as an official Department		
engineering each materia spectroscopy	organic framewo particles for use al using powder y, and nitrogen p	in toxic chemi X-ray diffraction porosimetry. Ne	on, attenuated total reflec either material showed sig	. Materials char tance – Fourier gns of structura	racterization was conducted on		
15. SUBJEC MOF; Cu	T TERMS 13(BTC)2; HKUS	ST-1; UiO-66;	Particle				
16. SECURIT	Y CLASSIFICAT	ON OF:	17. LIMITATION OF	15. NUMBE			
a. REPORT	b. ABSTRACT	c. THIS PAGE	ABSTRACT	OF PAGES	Krista Walton		
UU	UU	υυ	UU		19b. TELEPHONE NUMBER 404-894-5254		

Report Title

Effects of pelletization pressure on the physical and chemical properties of the metal–organic frameworks Cu3(BTC)2 and UiO-66

ABSTRACT

The metal—organic frameworks CuBTC and UiO-66 were pressed at 1000 and 10,000 psi as a first step to engineering particles for use in toxic chemical removal applications. Materials characterization was conducted on each material using powder X-ray diffraction, attenuated total reflectance – Fourier transform infrared spectroscopy, and nitrogen porosimetry. Neither material showed signs of structural degradation during pressing. The CuBTC pressed materials show reduced porosity after pressing, while the UiO-66 surface area remained consistent for all pressed samples. Small-scale breakthrough testing was conducted with CuBTC against ammonia, which probes the reactive sites, and UiO-66 against octane, which probes physical adsorption capacity. Even with the decrease in surface area, the CuBTC materials had consistent ammonia removal capacities, while the UiO-66 pressed materials showed a slight decrease in octane loadings. Data indicate that pressing these MOFs into pellets without a binder is a viable approach to engineering particles in support of filtration and other applications.

REPORT DOCUMENTATION PAGE (SF298) (Continuation Sheet)

Continuation for Block 13

ARO Report Number 57994.25-CH

Effects of pelletization pressure on the physical

Block 13: Supplementary Note

© 2013 . Published in Microporous and Mesoporous Materials, Vol. Ed. 0 179, (0) (2013), (, (0). DoD Components reserve a royalty-free, nonexclusive and irrevocable right to reproduce, publish, or otherwise use the work for Federal purposes, and to authroize others to do so (DODGARS §32.36). The views, opinions and/or findings contained in this report are those of the author(s) and should not be construed as an official Department of the Army position, policy or decision, unless so designated by other documentation.

Approved for public release; distribution is unlimited.

FISEVIER

Contents lists available at SciVerse ScienceDirect

Microporous and Mesoporous Materials

journal homepage: www.elsevier.com/locate/micromeso



Effects of pelletization pressure on the physical and chemical properties of the metal–organic frameworks Cu₃(BTC)₂ and UiO-66



Gregory W. Peterson a,*, Jared B. DeCoste b, T. Grant Glover b, Yougui Huang c, Himanshu Jasuja c, Krista S. Walton c

- ^a Edgewood Chemical Biological Center, 5183 Blackhawk Rd., Aberdeen Proving Ground, MD 21010, USA
- ^b SAIC, Gunpowder, MD 21010, USA
- ^cGeorgia Institute of Technology, School of Chemical & Biomolecular Engineering, Atlanta, GA 30332, USA

ARTICLE INFO

Article history: Received 11 January 2013 Received in revised form 16 February 2013 Accepted 18 February 2013 Available online 26 February 2013

Keywords: MOF Cu₃(BTC)₂ HKUST-1 UiO-66 Particle

ABSTRACT

The metal-organic frameworks CuBTC and UiO-66 were pressed at 1000 and 10,000 psi as a first step to engineering particles for use in toxic chemical removal applications. Materials characterization was conducted on each material using powder X-ray diffraction, attenuated total reflectance – Fourier transform infrared spectroscopy, and nitrogen porosimetry. Neither material showed signs of structural degradation during pressing. The CuBTC pressed materials show reduced porosity after pressing, while the UiO-66 surface area remained consistent for all pressed samples. Small-scale breakthrough testing was conducted with CuBTC against ammonia, which probes the reactive sites, and UiO-66 against octane, which probes physical adsorption capacity. Even with the decrease in surface area, the CuBTC materials had consistent ammonia removal capacities, while the UiO-66 pressed materials showed a slight decrease in octane loadings. Data indicate that pressing these MOFs into pellets without a binder is a viable approach to engineering particles in support of filtration and other applications.

© 2013 Published by Elsevier Inc.

1. Introduction

Sorbent design is an active field of study in a variety of commodity areas, including catalysis [1,2], gas storage [3,4], and separations [5,6]. Each of these areas requires specific micro- and macro-level properties. For example, chemical storage or removal may depend on a combination of physical properties, such as porosity and surface area, as well as chemical properties, such as reaction with surface groups. In all cases, materials must also be used in forms suitable for their host application, and engineered such that performance and energy (pressure drop) are optimized.

One of the more challenging applications for sorbent design is air purification and filtration. Due to the wide variety of chemicals that can be encountered, including highly toxic nerve agents, acidic/acid-forming gases such as chlorine, phosgene, sulfur dioxide, and hydrogen chloride, as well as basic gases such as ammonia, materials must be developed with several different functionalities capable of removing different classes of chemicals.

Metal-organic frameworks (MOFs) have been widely investigated over the past decade for use in these different applications [7–11]. MOFs combine metal nodes with organic linkers, forming extended nanoporous networks. They are of particular interest

due to the wide variety of functionalities that can be combined to tune physical and chemical properties of the sorbent. One such material, UiO-66, is a promising MOF for several reasons. It is one of the most stable MOFs known, as it does not degrade in acids, many typical solvents, or due to moisture [12]. It can also be functionalized in a variety of ways, either during initial synthesis by using linkers with pendant groups [13–16], or post-synthesis using chemical reactions or exchange [17,18]. Because of these properties, UiO-66 is essentially a molecular scaffold that can be tuned depending on the application.

Another promising material, CuBTC (aka Cu₃(BTC)₂, or HKUST-1), has been extensively studied in a variety of applications [19,20]. Although this material is not stable to moisture [21], methods have been developed to stabilize the material, such as with a plasma-enhanced chemical vapor deposition of perfluorohexane [22]. Peterson and coworkers demonstrated the high efficiency of CuBTC for ammonia removal [23], and therefore this MOF is a prime candidate for incorporation into filtration applications.

Several previous efforts have examined pelletizing MOFs [24–27], including UiO-66 and CuBTC [12,28]; however, many involve pelletization with a binder [29,30], which inherently changes the properties of the material. Many of the efforts also fail to examine the changes to porosity, density, and other important engineering parameters [12], or investigate liquid sorption [28] as opposed to toxic chemical removal. In addition to binders, it is common to

^{*} Corresponding author. Tel.: +1 (410) 436 9794.

E-mail address: gregory.w.peterson.civ@mail.mil (G.W. Peterson).

form pellets through the application of direct pressure on powders. The effect of pressure for forming MOF pellets on toxic chemical filtration capacity has not been extensively detailed in the literature. One recent effort by Kim and coworkers did in fact investigate using pressure with and without binders to create pellets of CuBTC [31]; however, X-ray diffraction data indicated the degradation of the structure at higher pressures. Moggach and coworkers investigated changes to the CuBTC crystal structure upon pressurization [32], while Cheetham and Tan reviewed the effect of pressure on variety of MOFs, including CuBTC [33].

This paper examines the first steps in materials engineering, with the objective of determining the effects of pressure during pelletization on the physical properties and chemical removal capabilities. UiO-66 and CuBTC were pressed at 1000 and 10,000 psi, followed by structural characterization and breakthrough testing.

2. Experimental

2.1. Materials synthesis

CuBTC was purchased from Sigma Aldrich, while UiO-66 was synthesized in-house. For the latter, 19.068 mmoles of zirconium (IV) chloride and 19.068 mmoles of benzene dicarboxyalate were mixed in 742 ml of dimethyl formamide (DMF) at room temperature in a glass beaker. The resulting mixture was divided in equal parts into three 500 ml glass jars. The jars were placed in a preheated oven at 120 °C for 24 h. The solution was cooled to room temperate, and the resulting solid was repeatedly washed with DMF. DMF was subsequently exchanged with methanol and then activated in a vacuum oven prior to forming particles.

2.2. Pellet and materials preparation

Pellets were prepared using a Carver Press. For UiO-66, two grams of powder were inserted into a 35 mm die and pressed at 1000 and 10,000 psi for one minute (UiO-66-P1000 and UiO-66-P10000, respectively). For CuBTC, two grams were loaded into the 35 mm die within a glove box, and then sealed in a plastic bag void of gases. The powder was pressed at 1000 and 10,000 psi for one minute within the plastic bag and removed within the glove box (CuBTC-P1000 and CuBTC-P10000 respectively). This procedure was conducted to keep moisture from destroying the crystal structure and porosity [21]. Each pellet was ground using a mortar and pestle within the plastic bag to make powders for breakthrough testing. In the case of CuBTC-P1000, the sample did not form a full pellet, and powder was still present in the die, indicating the pressure is not sufficient to provide an engineered material: nonetheless, the material was still evaluated for physical and chemical removal properties. In the materials pressed at higher pressures, materials were crushed to compare to virgin powder; however, it is possible to make engineered particles via crushing and sieving operations.

2.3. PXRD

Powder X-ray diffraction (PXRD) measurements were taken using a PANalytical X'Pert X-ray powder diffractometer with an X'celerator detector. Samples were scanned at 45 kV and 40 mA, using CuK α radiation (λ = 1.54 Å), a step size of 2θ = 0.033 $^{\circ}$ (10.08 s/step) over the 2θ range of 10–80 $^{\circ}$. Zero-background discs were used to minimize background scattering. PXRD patterns were processed using the Reflex module in Material Studio 5.5 by Accelrys.

2.4. Attenuated total reflectance – fourier transform infrared spectroscopy

Infrared transmission spectra were collected for samples using a Bruker Tensor 27 FTIR equipped with a Bruker Platinum attenuated total reflectance accessory with a single reflection diamond crystal. Scans were taken from approximately 4000 to 600 cm⁻¹. Sixteen scans were averaged, and resulting spectra were background subtracted and baseline corrected.

2.5. Nitrogen isotherm

Nitrogen porosimetry data were collected using a Quanta-chrome Autosorb 1C Analyzer. The UiO-66 samples were outgassed at $350\,^{\circ}\text{C}$ for approximately 24 h, and the CuBTC samples were outgassed at $150\,^{\circ}\text{C}$ for approximately 24 h. The surface area was calculated using the BET method [34], and the pore size distribution was calculated with non-local density functional theory (NLDFT) using a slit-pore carbon kernel. Total pore volume was calculated at a P/P₀ of 0.99, and the micropore volume was calculated using the Dubinin–Radushkevitch (DR) method.

2.6. Breakthrough testing

Breakthrough testing was conducted to assess the effect of pressure during pelletization on chemical removal performance. UiO-66 samples were evaluated against octane, while CuBTC samples were evaluated against ammonia. All samples were first activated at approximately 150 °C (CuBTC) to 170 °C (UiO-66) under dry flowing air for 1 h, and then loaded into 4 mm i.d. fritted glass tubes to a depth of 4 mm. Table 1 summarizes the test conditions. The ammonia microbreakthrough apparatus has been described previously [11,23]. Briefly, a ballast was pressurized with a known amount of ammonia, then mixed with a diluent stream at 0% relative humidity (RH) and atmospheric pressure at rates necessary to achieve a challenge concentration of 1,000 mg/m³. The stream then passed through the fritted glass tube packed with sorbent. The effluent concentration was continuously monitored using a Hewlett Packard (HP) 5890 gas chromatograph equipped with a photoionization detector (PID). Octane testing was conducted in a similar fashion, with the exception of the feed delivery and effluent monitoring. Dry air was passed over octane in a saturator cell, creating a saturated vapor stream, and then mixed with a diluent stream to achieve a challenge concentration of 4,000 mg/m³. Effluent monitoring was conducted with an HP GC equipped with a flame ionization detector (FID). The microbreakthrough system has an approximately standard deviation of 10%. Results are plotted on a weighted basis (min/g) to account for differences in density. Chemical loadings were calculated by integration of the area under the breakthrough curves.

Table 1 Microbreakthrough testing conditions.

Parameter	Ammonia system	Octane system
Temperature RH	20 °C 0% (-40 °C dew point)	20 °C 0% (-40 °C dew point)
Adsorbent mass Adsorbent volume	~10-25 mg 55 mm ³	~10-25 mg 55 mm ³
Flow rate Challenge concentration	20 mL/min 1000 mg/m ³	20 mL/min 4000 mg/m ³
chancinge concentration	1000 1118/111	1000 1118/111

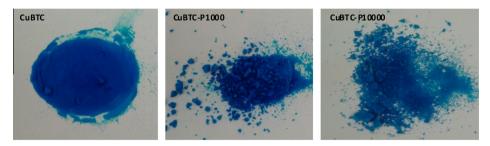


Fig. 1. CuBTC samples as-received, pelletized at 1000 psi, and pelletized at 10,000 psi. After pelletization, samples were crushed into small sizes to characterize.

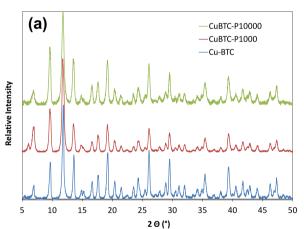
Table 2Surface area and pore volume measurements for CuBTC samples.

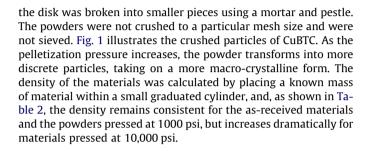
Sample	BET surface area (m²/g)	Total pore volume (cc/g)	%Micropore volume (%)	Density (cc/g)
CuBTC	1698	0.75	93.7	0.30
CuBTC-P1000	1045	0.61	73.2	0.28
CuBTC-P10000	892	0.52	73.3	0.41

3. Results and discussion

3.1. CuBTC pelletization

The use of the die resulted in a disk, or pieces of a disk, approximately 35 mm in diameter. In order to characterize the materials





3.1.1. Physical properties characterization

Fig. 2 shows the powder X-ray diffraction data collected to ensure that the crystal structure remained intact after pelletization. Overall, the patterns are consistent for the three preparations, indicating that the crystal structures remain intact. However, the sig-

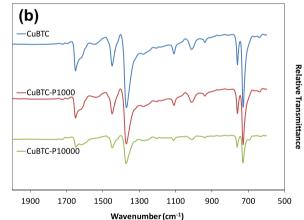


Fig. 2. (a) PXRD and (b) FTIR spectra for CuBTC samples.

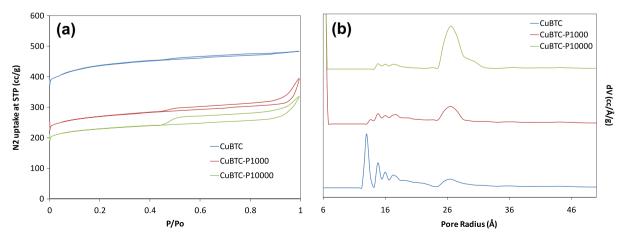


Fig. 3. (a) Nitrogen isotherms and (b) pore size distributions for CuBTC samples. As pelletization pressure increases, mesopores are formed within the material.

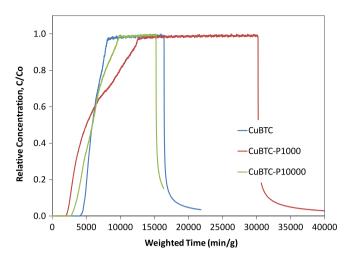


Fig. 4. Ammonia breakthrough results for CuBTC samples. Results show consistent ammonia removal for all three samples.

Table 3 Ammonia breakthrough results.

Sample	Loading (mmol/g)
CuBTC	7.2
CuBTC-P1000	7.2
CuBTC-P10000	7.1

nal-to-noise ratio decreases as the pressure increases, indicating at least partial structural collapse. To further ensure that the only changes to the powders were physical, and not chemical, FTIR spectra were collected, and are also shown in Fig. 2. The CuBTC exhibits the same spectra for all three preparations, indicating they remain structurally intact.

Nitrogen isotherms were collected to determine if porosity and pore size distribution were affected by the pelletization process. Fig. 3 illustrates the nitrogen uptake for the CuBTC samples, and Table 2 summarizes the BET surface area and pore volume of the three CuBTC samples. The as-received CuBTC exhibits a surface area consistent with the literature [35]. The as-received sample has the highest uptake, followed by the sample pressed at 1000 psi and finally the sample pressed at 10,000 psi. The decrease in surface area may be explained by localized pore structure collapse, which is supported by the reduced signal-to-noise of the CuBTC-P1000 sample. This is further supported by the development of mesopores, as seen by the hysteresis curves for the CuBTC-P1000 and especially the CuBTC-P10000 samples. Note the decrease in micropores and the increase in mesopores in the distribution plot of Fig. 3. It is possible that during the pelletization process, some bonds are broken, creating the mesopores.

Table 4Surface area and pore volumes for UiO-66 samples.

Sample	BET surface area (m²/g)	Total pore volume (cc/g)	%Micropore volume (%)	Density (cc/g)
UiO-66 UiO-66-P1000 UiO-66- P10000	1080 1080 1090	0.65 0.66 0.59	67.7 63.3 69.5	0.28 0.28 0.37

3.1.2. Ammonia microbreakthrough testing

CuBTC has previously been shown to provide excellent ammonia removal capabilities [23], and therefore samples were evaluated for their ammonia removal capabilities using a powder breakthrough system to determine the effects of pressing on performance. Fig. 4 illustrates the ammonia breakthrough curves. Results are plotted on a weighted basis to account for changes in density of the material. For both the pressed samples, ammonia begins eluting faster than the CuBTC as-received sample, possibly indicating issues of ammonia diffusing into the pores. The decreased sharpness of the breakthrough curves further supports diffusion resistance in the pressed samples. The total loading, as shown in Table 3, however, shows a consistent loading to saturation for all three samples. This indicates that, although the surface area and pore volume decrease with increasing pressure, the sites responsible for ammonia sorption are still accessible and active. Also of note is the sharp desorption curves, indicating ammonia is irreversibly adsorbed/reacted on the surface. The data shows that although the porosity decreases, pressing CuBTC into pellets does not degrade the total capacity of the material.

3.2. UiO-66 pelletization

Fig. 5 shows the crushed UiO-66 pellets, and Table 4 summarizes the surface area, pore volume, and densities of the materials. Note that, especially for the UiO-66-P10000 sample, the material is quite hard and retains significant macrostructure after crushing. Only the smallest granules were forwarded for octane testing. The density of the material also increases for the UiO-66-P10000 sample as compared to the UiO-66 material; however, the UiO-66-P1000 has a similar density to UiO-66.

3.2.1. Physical property characterization

Fig. 6 illustrates the PXRD data collected to determine changes to the crystal structure. It is apparent from the data that pressing the UiO-66 results in no change to the crystal structure. FTIR spectra indicate consistent functional groups between the samples, further indicating no chemical changes occurred during pressing.

Nitrogen isotherm data were collected to determine if the pelletization process affected the pore structure as is the case with CuBTC. Fig. 7 summarizes the nitrogen uptake and pore size distributions for the UiO-66 samples. Isotherm data for the samples indicate that pressure has little effect on the overall nitrogen up-







Fig. 5. UiO-66 samples as-received, pelletized at 1000 psi, and pelletized at 10,000 psi. After pelletization, samples were crushed into small sizes to characterize.

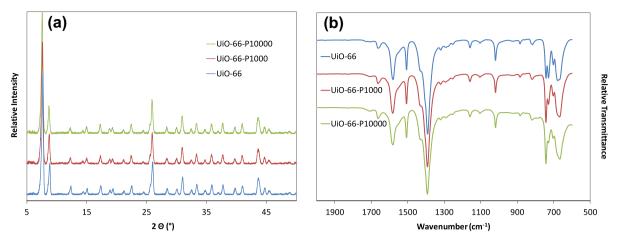


Fig. 6. (a) PXRD and (b) FTIR spectra for UiO-66 samples. Spectra for both techniques indicate no changes to the crystal structure or functional groups.

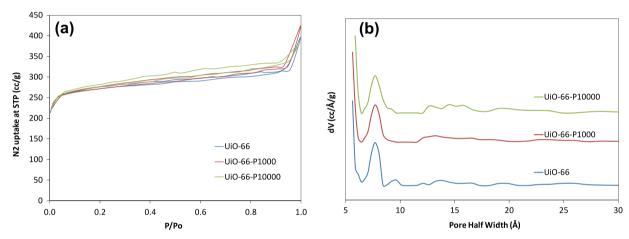


Fig. 7. (a) Nitrogen isotherm data and (b) pore size distributions for UiO-66 samples. Data indicate little effect of pressure on nitrogen uptake, but some increase in mesoporosity due to pressing.

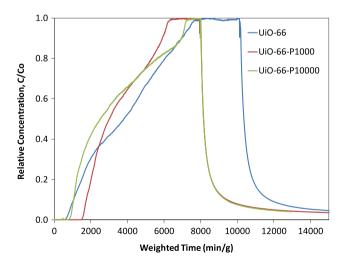


Fig. 8. Octane breakthrough results. The three samples tested show similar breakthrough behavior, indicating pelletization has limited effects on performance.

take, as all three isotherms are consistent throughout all relative pressures, within experimental error. The surface areas are also consistent for the three samples, although they are lower than previously reported values [36]. A reason for this is unknown, as the

Table 5Octane breakthrough results.

Sample	Loading (mmol/g)
UiO-66	2.5
UiO-66-P1000	2.3
UiO-66-P10000	2.1

crystal structure is consistent with reported X-ray patterns; therefore, it may be due to activation procedure. The pore size distributions are also fairly consistent, although the powder UiO-66 sample shows slightly higher microporosity, with mesoporosity developing in the pressed samples. This behavior can be expected, as larger pores may be developed in between the crystals of the pressed structures during pelletization.

3.2.2. Octane microbreakthrough testing

UiO-66 has limited functionality, and therefore octane was chosen as a test gas to evaluate the effects of pressure on performance. Powders were tested at a constant volume, with the mass varying according to changes in density from the pelletization process. Fig. 8 illustrates the results, which are plotted on a weighted time and relative concentration basis. Three replicates were tested for each sample. Surprisingly, the baseline UiO-66 sample exhibits octane breakthrough prior to the two pressed samples. However, all

three samples reach saturation at approximately the same weighted time. And, as shown in Table 5, the UiO-66 sample actually has the highest octane adsorption capacity, with capacity decreasing as pelletization pressure increases. Even after pressing at 10,000 psi, however, the material still retains significant octane removal capacity. These results are consistent with the nitrogen isotherm data, which show little or no change in porosity and surface area of the materials after pelletization.

4. Conclusions

The metal–organic frameworks CuBTC and UiO-66 were pressed at 1000 and 10,000 psi as a first step to engineering particles for use in chemical removal applications. In both cases, PXRD and FTIR data indicate that the crystal structures remain intact after pressing. The surface area and porosity of CuBTC decrease significantly after pressing, whereas the pressure has much less of an effect on UiO-66. Although the CuBTC material exhibited reduced porosity, the ammonia removal performance was unaffected, indicating that engineering CuBTC into hard granules is possible without detrimental effects. Octane breakthrough results do indicate degradation in performance for the UiO-66 material, but even the sample pressed at 10,000 psi still retains significant removal capacity.

Results from the study indicate that indeed both CuBTC and UiO-66 can be engineered into forms suitable for chemical removal applications. Although some degradation in porosity is seen for the samples, both still exhibit substantial chemical removal capabilities after pelletization. Further studies will be conducted to optimize hardness and density for larger-scale testing.

Acknowledgements

This work was completed through funding by the Defense Threat Reduction Agency under Project BA07PRO104 (ECBC) and ARO Contract Number W911NF-10-0076 (KSW). The authors thank Matthew Browe and Paulette Jones for conducting the breakthrough testing.

References

- S. Sato, Y. Yoshihiro, Y. Hidenori, M. Noritaka, M. Iwamoto, Applied Catalysis 70 (1991) L1–L5.
- [2] A. Corma, M.S. Grande, V. Gonzalez-Alfaro, A.V. Orchilles, J. Catal. 159 (1996) 375–382.
- [3] J.L.C. Rowsell, O.M. Yaghi, J. Am. Chem. Soc. 128 (2006) 1304-1315.
- [4] Q. Yang, C. Zhong, J. Phys. Chem. B 110 (2006) 655–658.

- [5] J. Padin, R.T. Yang, Chem. Eng. Sci. 55 (2000) 2607-2616.
- [6] X. Xiaochun, S. Chunshan, B.G. Miller, A.W. Scaroni, Ind. Eng. Chem. Res. 44 (2005) 8113–8119.
- [7] T. Duren, L. Sarkisov, O.M. Yaghi, R.Q. Snurr, Langmuir 20 (2004) 2683– 2689.
- [8] A.R. Millward, O.M. Yaghi, J. Am. Chem. Soc. 127 (2005) 17998-17999.
- [9] J.G. Vitillo, L. Regli, S. Chava, G. Ricchiardi, G. Spoto, P.D.C. Dietzel, S. Bordiga, A. Zecchina, J. Am. Chem. Soc. 130 (2008) 8386–8396.
- [10] A.G. Wong-Foy, A.J. Matzger, O.M. Yaghi, J. Am. Chem. Soc. 128 (2006) 3493–3495.
- [11] T.G. Glover, G.W. Peterson, B.J. Schindler, D. Britt, O. Yaghi, Chem. Eng. Sci. 66 (2011) 163–170.
- [12] J.H. Cavka, S. Jakobsen, U. Olsby, N. Guillou, C. Lamberti, S. Boriga, K.P. Lillerud, J. Am. Chem. Soc. 130 (2008) 13850–13851.
- [13] M. Kandiah, M.H. Nilsen, S. Usseglio, S. Jakobsen, U. Olsbye, M. Tilset, C. Larabi, E.A. Quadrelli, F. Bonino, K.P. Lillerud, Chem. Mater. 22 (2010) 6632– 6640.
- [14] Y. Huang, W. Qin, Z. Li, Y. Li, Dalton Trans. 41 (2012) 9283-9285.
- [15] G.E. Cmarik, M. Kim, S.M. Cohen, K.S. Walton, Langmuir 28 (2012) 15606– 15613.
- [16] H. Jasuja, J. Zang, D. Sholl, K.S. Walton, J. Phys. Chem. C 116 (2012) 23526– 23532.
- [17] S.J. Garibay, S.M. Cohen, Chem. Comm. 46 (2010) 770-7702.
- [18] M. Kandiah, S. Usseglio, S. Svelle, U. Olsbye, K.P. Lillerud, M. Tilset, J. Mater. Chem. 20 (2010) 9848–9851.
- [19] S.S.Y. Chui, S.M.F. Lo, J.P.H. Charmant, A.G. Orpen, Science 283 (1999) 1148– 1150.
- [20] B. Xiao, P.S. Wheatley, X. Zhao, A.J. Fletcher, S. Fox, A.G. Rossi, I.L. Megson, S. Bordiga, L. Regli, K.M. Thomas, R.E. Morris, J. Am. Chem. Soc. 129 (2007) 1203–1209.
- [21] P.M. Schoenecker, C.G. Carson, H. Jasuja, C.J.J. Flemming, K.S. Walton, Ind. Eng. Chem. Res. 51 (2012) 6513–6519.
- [22] J.B. DeCoste, G.W. Peterson, M.W. Smith, C.A. Stone, C.R. Willis, J. Am. Chem. Soc. 134 (2012) 1486–1489.
- [23] G.W. Peterson, G.W. Wagner, A. Balboa, J. Mahle, T. Sewell, C.J. Karwacki, J. Phys. Chem. C 113 (2009) 13906–13917.
- [24] M. Tagliabue, C. Rizzo, R. Millini, P.D.C. Dietzel, R. Blom, S. Zanardi, J. Porous Mater. 18 (2011) 289–296.
- [25] J. Liu, J. Tian, P.K. Thallapally, B.P. McGrail, J. Phys. Chem. C 116 (2012) 9575– 9581
- [26] J. Liu, Y. Wang, A.I. Benin, P. Jakubczak, R.R. Willis, M.D. LeVan, Langmuir 26 (2010) 14301–14307.
- [27] J. Liu, P.K. Thallapally, D. Strachan, Langmuir 28 (2012) 11584-11589.
- [28] P.S. Barcia, D. Guimaraes, P.A.P. Mendes, J.A.C. Silva, V. Guillerm, H. Chevreau, C. Serre, A.E. Rodrigues, Microporous Mesoporous Mater. 139 (2011) 67–73.
- [29] V. Finsy, L. Ma, L. Alaerts, D.E. De Vos, G.V. Baron, J.F.M. Denayer, Microporous Mesoporous Mater. 120 (2009) 221–227.
- [30] L.D. O'Neill, H. Zhang, D. Bradshaw, J. Mater. Chem. 20 (2010) 5720-5726.
- [31] J. Kim, S.-H. Kim, S.-T. Yang, Microporous Mesoporous Mater. 161 (2012) 48–55.
- [32] A.J. Graham, J.-C. Tan, D.R. Ilan, S.A. Moggach, Chem. Commun. 48 (2012) 1535–1537.
- [33] J.C. Tan, A.K. Cheetham, Chem. Soc. Rev. 40 (2011) 1059-1080.
- [34] K.S. Walton, R.Q. Snurr, J. Am. Chem. Soc. 129 (2007) 8552–8556.
- [35] M. Hartmann, S. Kunz, D. Himsl, O. Tangermann, S. Ernst, A. Wagener, Langmuir 24 (2008) 8634–8642.
- [36] H.R. Abid, H. Tian, H.-M. Ang, M.O. Tade, C.E. Buckley, S. Wang, Chem. Eng. J. 187 (2012) 415–420.